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Synthesis, Structure and Properties of the \((\text{Bi}_{0.5}\text{Sr}_{0.5})\)(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) Perovskite

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Dedicated to Professor Gérard Demazeau on the occasion of his 65th birthday

The new perovskite \((\text{Bi}_{0.5}\text{Sr}_{0.5})\)(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) has been prepared under high pressure (10 GPa) and temperature (900 °C). The room-temperature crystal structure is described by the space group \(Pnma\) (\(a = 5.5702(3)\), \(b = 7.8793(5)\) and \(c = 5.5599(4)\) Å) with no observed order between Bi and Sr or between Co and Ru cations. This material shows antiferromagnetic spin ordering or freezing below 50 K and a paramagnetic moment of 4.07 \(\mu_B\) per formula unit at high temperatures, consistent with the presence of high-spin \(\text{Co}^{2+}\) or \(\text{Co}^{3+}\) ions. \((\text{Bi}_{0.5}\text{Sr}_{0.5})\)(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) is a variable range hopping semiconductor between 40 and 300 K with a small negative magnetoresistance.

Key words: Disordered Perovskite, Distorted Perovskite, High-pressure Synthesis, Antiferromagnetism

Introduction

Transition metal oxides of the perovskite-type structure show important physical properties including ferroelectricity and ferromagnetism. These oxides have a general formula \(\text{ABO}_3\) with \(\text{A}\) typically being an alkaline, alkaline earth or rare earth cation and \(\text{B}\) often a transition metal [1]. Interesting and useful properties can arise from the coupling of spin, charge and orbital degrees of freedom of the transition metal ions.

When two different ions are ordered on the \(B\) sites of the perovskite structure, an \(\text{A}_2\text{BB}'\text{O}_6\) double perovskite is obtained [2]. The possibility of different degrees of cation order-disorder in these oxides adds another variable that can play an important role in their properties. Oxides of this type have received much attention in recent years due to properties such as large magnetoresistance (e.g. \(\text{Sr}_2\text{FeMoO}_6\) [3]) and high ionic conductivity in \(\text{Sr}_2\text{Mg}_{1-x}\text{Mn}_x\text{MoO}_6\) [4]. Co-existing ferroelectricity and ferromagnetism (multiferroic behaviour) has been reported in \(\text{Bi}_2\text{NiMnO}_6\) [5], which may be useful in future electronic devices [6]. In such double perovskites, ferroelectricity can arise from off-centre displacements of a polar cation such as \(\text{Bi}^{3+}\) or \(\text{Pb}^{2+}\) in the \(\text{A}\) sites, while magnetic ordering may result from the exchange coupling of the transition metal cation spins at the \(B\) sites. However, \(B\) cations in similar oxidation states (as in the predicted multiferroic \(\text{Bi}_2\text{CrFeO}_6\) [7]) are usually disordered [8], and high-temperature and -pressure conditions are needed to stabilize \(\text{Bi}^{3+}\) in the \(A\) sites (typically \(\sim 6\) GPa and \(\sim 1000\) °C). Of the phases prepared recently, only \(\text{Bi}_2\text{NiMnO}_6\) is cation ordered whereas \(\text{BiCo}_{0.5}\text{Mn}_{0.5}\text{O}_3\) [5], \(\text{BiCu}_{0.5}\text{Mn}_{0.5}\text{O}_3\) [5] and \(\text{BiFe}_{0.5}\text{Cr}_{0.5}\text{O}_3\) [9] show partial or complete disorder and predominantly antiferromagnetic behaviour at low temperatures.

High-pressure and -temperature conditions have been used to prepare new materials since the 1970’s but the potential of this synthetic route has only been exploited heavily in the last decade. In the perovskite field, the use of high pressure has led to the discovery of complex \(A\) and \(B\) cation orderings, Bi-containing multiferroics, new manganites showing colossal magnetoresistances, and phases with elements in unusual oxidation states [10 – 12]. Following these previous works, we have attempted the synthesis of the \((\text{BiSr})\text{CoRuO}_6\) double perovskite under high pressure. Substantial \(\text{Co/Ru} B\) site ordering was previously observed in the double perovskites \((\text{LaCa})\text{CoRuO}_6\), \((\text{LaBa})\text{CoRuO}_6\), and \((\text{La}_{1+x}\text{Sr}_{1-x})\text{CoRuO}_6\) (\(-0.5 < x < 0.25\)) [13, 14].
Experimental Section

The polycrystalline sample was synthesised under high pressure using a Walker-type multianvil apparatus. The precursor was made by a wet-chemical route using a citric acid sol-gel. Stoichiometric quantities of high-purity SrCO₃, Co(CH₃COO)₂, Bi₂O₃ and RuO₂ were dissolved in dilute aqueous nitric acid. Citric acid (3 mol per mol of metal) was added to the aqueous solution which was stirred for 1 h to ensure complexation. Ethylene glycol was then added (3/4 mol per mol of citric acid). The solution was dehydrated in a sand bath at 100 °C to a gel which was slowly decomposed in a crucible over one day at 450 °C in air. The resulting mixed oxide powder was ground and calcined several times at temperatures up to 950 °C for 6 h.

∼ 20 mg portions of the precursor were transferred to a cylindrical boron nitride crucible at the centre of a pre-cast MgO octahedron, which was compressed to 10 GPa by eight truncated tungsten carbide cubes. A cylindrical graphite resistance heater contacted by two molybdenum plates was used to heat the sample, which was then cooled to ambient temperature crystal structure was analysed by the Rietveld method [15] using the GSAS software package [16].

After heating to 900 °C for 6 h, approximately 15 mg of an almost phase pure sample were recovered for the X-ray powder diffraction data of (Bi₀.₅Sr₀.₅)₅Ru₀.₅Co₀.₅O₃ at T = 300 K. The peak profile was fitted using a modified pseudo-Voigt function, and the background was fitted with a Ṽoigt function, and the background was fitted with a

<table>
<thead>
<tr>
<th>Atom</th>
<th>Site</th>
<th>x</th>
<th>y</th>
<th>z</th>
</tr>
</thead>
<tbody>
<tr>
<td>Sr</td>
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<td>0.028(3)</td>
<td>1/4</td>
<td>0.0090(7)</td>
</tr>
<tr>
<td>Ru</td>
<td>4b</td>
<td>0.003(3)</td>
<td>1/4</td>
<td>0.538(7)</td>
</tr>
<tr>
<td>O1</td>
<td>8d</td>
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<td>0.028(4)</td>
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</table>

Table 1. Structure refinement results for (Bi₀.₅Sr₀.₅)₅Co₀.₅Ru₀.₅O₃ from the Rietveld fit to the room-temperature X-ray diffraction data (space group Pnma; a = 5.5702(3), b = 7.8793(5), c = 5.5599(4) Å, V = 244.02(2) Å³, Z = 4; overall Uiso = 0.0078(4) Å²; residuals: Rp = 3.9 %, Rp = 2.9 %, Rp = 6.9 %. χ² = 1.9 %).

<table>
<thead>
<tr>
<th>Site</th>
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<th>b</th>
<th>c</th>
</tr>
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<tbody>
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<td>Co/Ru-O1</td>
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<td>Co/Ru-O2</td>
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<tr>
<td>O1-Co/Ru-O2</td>
<td>91(2)</td>
<td>Co/Ru-O2-Co/Ru</td>
<td>162(1)</td>
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<tr>
<td>O2-Co/Ru-O2</td>
<td>90(2)</td>
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</tbody>
</table>

Table 2. Selected bond lengths (Å) and angles (deg) for (Bi₀.₅Sr₀.₅)₅Co₀.₅Ru₀.₅O₃.

Fig. 1. Rietveld fit to the room-temperature X-ray powder diffraction data of (Bi₀.₅Sr₀.₅)₅Co₀.₅Ru₀.₅O₃ in space group Pnma, showing observed, calculated and difference curves. The fit to the high-angle data is inset. Lower tick marks correspond to ∼ 1% SrBiO₃ impurity (highest peak indicated as *).

Results

Crystal structure

The X-ray diffraction data show a large number of weak reflections which indicate a significant distortion of the cubic perovskite structure. All peaks were indexed by considering a √2ap × 2ap × √2ap expansion of the cubic lattice using space group Pnma (no. 62). This space group does not describe ordering on the A or B cation sites, showing that the product is the cation-disordered perovskite (Bi₀.₅Sr₀.₅)₅Co₀.₅Ru₀.₅O₃. Attempts to refine the structure in the monoclinic space group P2₁/n which allows B-site ordering confirmed the lack of order so the final refinements were carried out using the Pnma orthorhombic model.

The peak profile was fitted using a modified pseudo-Voigt function, and the background was fitted with a
linear interpolation function. Variable Bi/Sr and oxygen coordinates were refined, and the isotropic displacement parameters of all atoms were constrained to be equal. Final refined parameters and residuals are given in Table 1, and selected bond lengths and angles are shown in Table 2. The observed, calculated, and difference profiles are shown in Fig. 1.

The crystal structure of (Bi$_{0.5}$Sr$_{0.5}$)(Co$_{0.5}$Ru$_{0.5}$)O$_3$ is characterised by the tilting of the Co/RuO$_6$ octahedra along the three axes of the primitive perovskite cell, that can be described as $a^+b^-b^-$, according to the Glazer notation [17]. This changes the Co/Ru–O–Co/Ru angles from the ideal value of 180$^\circ$ to the observed range of 162 – 168$^\circ$ as shown in Table 2. However, the Co/RuO$_6$ octahedra remain almost undistorted with internal O–Co/Ru–O angles of 90–93$^\circ$.

The Co/Ru–O distances show an apparent tetragonal elongation of the Co/RuO$_6$ octahedra although this is not significant within the error limits. The mean Co/Ru–O distance (Table 2) of 2.00(4) Å is comparable to the expected values for the high-spin Co$^{2+}$–Ru$^{5+}$ (2.05 Å), high-spin Co$^{3+}$-low-spin Ru$^{4+}$ (2.01 Å) and low-spin Co$^{3+}$-low-spin Ru$^{4+}$ (1.97 Å) configurations, but is too imprecise to enable the cation oxidation states to be assigned. The mean Bi/Sr–O distance of 2.66(4) Å is close to the expected value of 2.61 Å. (The calculated distances use published ionic radii [18] assuming 8, 6, and 14/3 coordination numbers for the A, B and O atoms, to allow for the reduced coordination around the A cations in the distorted Pnma superstructure).

**Magnetic properties**

The temperature dependences of the molar magnetic susceptibility $\chi_m$ and the inverse susceptibility for (Bi$_{0.5}$Sr$_{0.5}$)(Co$_{0.5}$Ru$_{0.5}$)O$_3$ are shown in Fig. 2a. At high temperatures (> 160 K) $\chi_m$ follows a Curie-Weiss law, $\chi_m = C_m/(T - \theta)$, with $C_m$ = 2.06 cm$^3$·K mol$^{-1}$ and $\theta$ = −150 K, indicating that the predominant exchange interactions are antiferromagnetic. The derived high-temperature paramagnetic moment is 4.07 $\mu_B$ per (Bi$_{0.5}$Sr$_{0.5}$)(Co$_{0.5}$Ru$_{0.5}$)O$_3$ formula unit (f. u.). This value is within the range of 3.9–4.6 $\mu_B$ expected for the high-spin Co$^{2+}$–Ru$^{5+}$ combination, allowing for a possible orbital contribution to the Co$^{2+}$ moment, but
is also close to the spin-only estimate of 4.0 \( \mu_B \) for high-spin Co\(^{3+}\) - low-spin Ru\(^{4+}\). The low-spin Co\(^{3+}\) - low-spin Ru\(^{4+}\) combination would have an effective moment of 2.0 \( \mu_B \) and so is ruled out.

The broad ZFC susceptibility maximum observed for \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) at 50 K shows that antiferromagnetic spin ordering or freezing occurs at low temperatures. The ZFC and FC susceptibilities shown in Fig. 2b diverge below 50 K which could be due to weak ferromagnetism or a spin-glass behaviour. The magnetisation \(M\) vs. applied field loop at 5 K in \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) shows no significant hysteresis. The slight \(H_{\text{f}}\) offset of the linear \(M-H\) relation corresponds to a very small saturated moment of 0.002 \( \mu_B \) per f. u. which is more likely to be due to a trace of a ferromagnetic impurity such as \(\text{SrRuO}_3\) than intrinsic weak ferromagnetism in the main \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) phase. Altogether, the low temperature measurements suggest that at least some glassy character to the spin order is present. Neutron diffraction would be needed to establish clearly whether long-range spin order occurs below \(\sim 50\) K in \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\).

**Transport properties**

The temperature dependence of the electrical resistivity of \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) is shown in Fig. 3. The observed curve is consistent with semiconducting behaviour, and the low temperature data were fitted using the equation (1) derived from the Mott variable-range hopping (VRH) mechanism [19]:

\[
\rho = \rho_0 \sqrt{T/T_0} \exp(T_0/T)^{1/(d+1)}
\]

where \(d\) is the dimensionality of the hopping process and \(T_0\) is the characteristic temperature. The pre-factor is temperature dependent, but is often neglected because of the stronger temperature dependence of the exponential term. The linear variation of \(\log \rho\) vs. \(T^{-1/4}\) (Fig. 3, inset) indicates that a three-dimensional \((d = 3)\) VRH process accounts well for the semiconductivity at low temperatures.

The field dependence of the resistivity of \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) was measured at 150 and 300 K. Fig. 4 shows the observed magnetoresistance (MR), calculated as \(\text{MR}(H,T) = [\rho(H,T) - \rho(0,T)]/\rho(0,T)\) where \(H\) denotes the applied field and \(\rho(0,T)\) and \(\rho(H,T)\) are the resistivities at zero and \(H\) fields, respectively. \((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) shows no significant magnetoresistance at 300 K but a small negative magnetoresistance, up to 0.5 % at 70 kOe, is observed at 150 K.

**Discussion**

\((\text{Bi}_{0.5}\text{Sr}_{0.5})(\text{Co}_{0.5}\text{Ru}_{0.5})\text{O}_3\) shows a disordered arrangement of Co and Ru over the \(B\) sites of an orthorhombic \(Pnma\) perovskite structure. The lack of order in this type of complex perovskite shows that there is only a small difference between the oxidation states of the \(B\) cations or that they have similar sizes. Similar disorder is found in the phases \(\text{SrCo}_{0.5}\text{Ru}_{0.5}\text{O}_3\) [13]
and La_{0.5}Sr_{0.5}Mn_{0.5}Ru_{0.5}O_{3}, [20] with Co^{3+}–Ru^{5+} and Mn^{3+}–Ru^{4+} pairs, respectively. Partial disorder has been claimed for the M^{2+}–Ru^{5+} pairs in Ba_{0.5}Zn_{0.5}Ni_{0.5}Ru_{0.5}O_{3} and Ba_{0.5}La_{0.5}Ni_{0.5}Ru_{0.5}O_{3} [21–23], although a high degree of order has been observed in the similar series (La_{1-x}Sr_{x})CoRuO_{6} (−0.50 ≤ x ≤ 0.25), where the Co^{2+}–Ru^{5+} combination is observed at x = 0 [14]. The greater disruption of the oxygen positions through off-centre distortions of the Bi^{3+} cations in comparison to La^{3+} may suppress long range Co/Ru order in (Bi_{0.5}Sr_{0.5})(Co_{0.5}Ru_{0.5})O_{3}. The B-site disorder may also lead to a mixture of Co^{2+}/3+ and Ru^{4+}/5+ states to minimise the Coulombic repulsion energies. Neither the structural nor the magnetic data are able to distinguish between these two limiting configurations, although the magnetic data do demonstrate that the Co ions are in a high-spin state.

The observed structural disorder in (Bi_{0.5}Sr_{0.5})(Co_{0.5}Ru_{0.5})O_{3} leads to a wide range of possible superexchange interactions between the paramagnetic ions and complicates the interpretation of the magnetic data. The predominant interactions are evidently due to antiferromagnetic M–O–M superexchange, as evidenced by the large negative value of the Weiss temperature, θ = −150 K. The divergence observed between the ZFC and FC curves below the 50 K maximum in the ZFC susceptibility shows that there is a significant glassy component to the spin order. This could be a frozen partial disorder within an antiferromagnetic state, or the ground state of (Bi_{0.5}Sr_{0.5})(Co_{0.5}Ru_{0.5})O_{3} may be a spin glass without any long-range magnetic order.

No resistive anomaly is observed at the 50 K spin freezing or ordering transition, and (Bi_{0.5}Sr_{0.5})(Co_{0.5}Ru_{0.5})O_{3} is semiconducting with 3-dimensional variable range hopping behaviour at low temperatures. A small magnetoresistance is observed in the paramagnetic region, and the linear variation of MR with H shows that it is intrinsic rather than a grain boundary effect. The magnetoresistance may result from local ferromagnetic order and spin-polarised hopping, as is found in several doped cobalt oxides.

In summary, the new phase (Bi_{0.5}Sr_{0.5})(Co_{0.5}Ru_{0.5})O_{3} has been prepared at 10 GPa and 900 °C. It adopts a distorted perovskite-type crystal structure described by the space group Pnma with no observed order between the Co and Ru cations. The B-site charge distribution may be Co^{2+}/5+/Ru^{5+} or Co^{3+}/Ru^{4+} or intermediate between these two combinations, with Co ions in the high-spin state. Spin ordering or freezing occurs below 50 K. (Bi_{0.5}Sr_{0.5})(Co_{0.5}Ru_{0.5})O_{3} is a variable range hopping semiconductor showing a small negative magnetoresistance.

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