Cation, magnetic, and charge ordering in MnFe3O5

Citation for published version:

Digital Object Identifier (DOI): 10.1039/C8TC00053K

Link:
Link to publication record in Edinburgh Research Explorer

Document Version:
Peer reviewed version

Published In:
Journal of Materials Chemistry C

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Introduction
The variable 3d configurations of Mn and Fe cations in spinel-type and other oxides give rise to many important applications as energy materials and in magnetism. Many manganese oxides are used as cathode materials in rechargeable batteries, e.g. LiMnO₂-spinel, Li₂MnO₃, and NaMnO₂. The phosphate LiFePO₄ has been commercialised as a lithium battery cathode and iron oxides such as Fe₂O₃ and magnetite, Fe₃O₄, have been explored as anode materials. Manganese and iron based spinels are also active catalysts for oxygen reduction/evolution reactions (ORR/OER) in fuel cells, metal-air batteries, and water-splitting cells.

The spinel magnetite is also notable as the original magnetic material, and Fe₃O₄ has been studied intensively since Verwey’s observation of a metal-insulator transition accompanied by a structural distortion. The low temperature structure has a complex charge and orbital ordering and weak Fe-Fe bonding interactions that form trimerons - linear orbital molecule clusters of three Fe ions. A related iron oxide, Fe₆O₈, was recently discovered using high temperature and high pressure synthesis, and has an incommensurate charge order at 150 K, below which dimeron and trimeron-like groups of Fe ions are formed. Subsequent work has shown that Fe₂O₄ₙ₋₁ homologues with n > 4 can also be made at pressure. MnFe₂O₄ analogues of these materials with Mn³⁺ = Ca were reported previously, and recent detailed studies of CaFe₂O₄ revealed a coupled structural and magnetic transition at 360 K accompanied by Fe²⁺/Fe³⁺ charge ordering.

We recently reported the synthesis at high pressures of the first n > 3 Mn material in this family, the n = 4 member MnFe₄O₇. MnFe₂O₄ is isostructural with Fe₄O₅ and adopts the orthorhombic Sr₂Ti₂O₇-type structure (space group Cmcm) in which divalent cations occupy triangular prismatic sites within triangular channels in a network of corner and edge-sharing octahedra. MnFe₂O₄ showed two magnetic transitions - an antiferromagnetic transition at 350 K and a broad ferromagnetic transition at 150 K – indicating that complex spin-spin interactions are present. We report here a high resolution powder neutron diffraction study of MnFe₂O₄, revealing the Mn/Fe chemical order and the temperature evolution of the magnetic behaviour between 5 and 400 K, supported by low temperature powder synchrotron X-ray diffraction and the electrical resistivity measurements.

Experimental
MnFe₂O₄ was synthesised using high pressure and high temperature solid state synthesis. Powders of MnO and Fe₂O₃ were ground together in a 1:1 ratio, and were heated at 1400 °C in a Pt capsule for 20 min under 10 GPa pressure, using a two-stage Walker-type module. Products were characterised by laboratory X-ray diffraction data collected with a Bruker D2 diffractometer using Cu-Kα radiation.

Magnetic measurements were carried out with a Quantum Design MPMS XL SQUID magnetometer. Magnetic susceptibility was recorded in zero field cooled (ZFC) and field cooled (FC) conditions between 2 and 400 K with an applied magnetic field of 5000 Oe. Hysteresis loops were measured at 2, 75, 300 and 400 K. Electrical resistivity measurements were carried out with a Quantum Design PPMS, between 260 and 380 K.

High resolution time-of-flight neutron diffraction data were collected at the WISH beamline of the ISIS facility, with 50 mg of powder from several high pressure runs packed into a vanadium can. Diffraction patterns were collected at 5, 75, 150, 300 and 400 K using a closed cycle refrigerator (CCR) with a hot stage. High resolution powder X-ray diffraction data were

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Cation, magnetic, and charge ordering in MnFe₂O₄†

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The recently-discovered high pressure material MnFe₂O₄ displays a rich variety of magnetically ordered states on cooling. Fe spins order antiferromagnetically below a Néel transition at 350 K. A second transition at 150 K marks Mn spin order that leads to spin canting of some of the Fe spins and ferrimagnetism. A further transition at 60 K is driven by charge ordering of Fe²⁺ and Fe³⁺ over two inequivalent Fe sites, with further canting of all spins. Electrical resistivity measurements reveal semiconducting behaviour in MnFe₂O₄ with a change in activation energy at 285 K.

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§Electronic Supplementary Information (ESI) available: Tables of crystallographic results and powder synchroton profile fit. See DOI: 10.1039/x0xx00000x. Open data for this article are at http://datashare.is.ed.ac.uk/handle/10283/838.
collected at the ID22 beamline of the ESRF with incident wavelength 0.39994 Å. A glass capillary with an outer diameter of 0.3 mm was used to contain the polycrystalline sample. Low temperature diffraction data were collected from 13 to 120 K using a liquid helium cryostat system.

Results and discussion

Crystal and magnetic structures

High resolution time-of-flight neutron diffraction experiments were carried out to determine the structure and magnetic behaviour of MnFe$_3$O$_5$. Crystal structure refinements show that the unit cell symmetry remains orthorhombic Cmcm at all temperatures. Occupancy refinements for the three cation sites, making use of the high neutron scattering contrast between Mn and Fe, showed that octahedral sites are occupied exclusively by Fe, and the trigonal prism site is predominantly Mn with 6.3(4) % substitution by Fe. Hence MnFe$_3$O$_5$ is Mn/Fe ordered and the overall Mn$_{0.94}$Fe$_{3.06}$O$_5$ composition of the present sample is close to the ideal stoichiometry.

Magnetisation measurements (Figure 1) show a maximum at 350 K and a broad magnetic upturn signifying a ferro- or ferrimagnetic ordering at 150 K as reported previously. This is corroborated by the saturated magnetisations observed in hysteresis loops at 2 and 75 K.

The neutron diffraction patterns shown in Figure 2 reveal the appearance of magnetic reflections when cooled below 400 K. The magnetic structures in the different regimes were solved and Rietveld fitted to the neutron diffraction patterns, as shown for the 5 K pattern in Figure 3. The magnetic reflections from all the magnetic phases of MnFe$_3$O$_5$ were indexed with a propagation vector of (000), and the structures obtained at 5, 75 and 300 K are presented in Figure 4a. At 150 and 300 K, the spins at the two independent Fe sites in MnFe$_3$O$_5$ are both found to be ordered antiferromagnetically parallel to the c-axis, whilst the Mn spins remain disordered. This confirms that a Néel transition is observed at $T_N = 350$ K in the magnetisation measurements. (Fig. 1)

Fits to the 75 K diffraction data showed additional spin ordering of the Mn site, with moments aligned ferromagnetically along the $b$ axis below $T_{MN} = 150$ K. The order of the Mn site moment leads the spins of the nearest Fe site - Fe1 to cant towards the $b$ axis. The $b$-components of Fe1 spins are antiparallel to those of Mn, resulting in a net magnetisation of ~0.5 $\mu_B$ per MnFe$_3$O$_5$ formula unit, which is consistent with the increase in magnetisation on cooling from 300 to 75 K shown in Fig. 1. A similar spin canting was reported in Fe$_3$O$_5$.8
The additional magnetic reflections observed in the 5 K neutron diffraction patterns reveal another change in the magnetic structure in MnFe$_2$O$_5$. The onset for this third magnetic transition appears to be the divergence between zero-field cooled and field cooled susceptibilities at ~60 K in the susceptibility data (Figure 1). An increase in the ordered Mn moment leads to further canting of all the spins, with both of the Fe sites canted antiferromagnetically towards the $a$ axis and ferromagnetically along $b$. In addition, the Mn spins become canted antiferromagnetically to the $c$ axis. The magnetic components of the Mn and the Fe2 sites on the $b$ axis are antiparallel to Fe1, as shown in Fig. 4a. This enhances the magnetisation along the $b$ axis to ~0.6 $\mu_B$ per MnFe$_2$O$_5$ formula unit. The thermal evolution of the magnitude of the ordered moments in MnFe$_2$O$_5$ are shown in Figure 4b, and the values of the ordered components and other refinement results are given in Supplementary Information.

The 300 K magnetic structure reveals dominant antiferromagnetic Fe1-Fe2 interactions; from direct exchange via overlap of half-filled $t_{2g}$ orbitals through edge-sharing of FeO$_6$ octahedra, and through superexchange mediated by Fe-O-Fe connections at shared corners. Each Mn spin is coupled to 4 Fe1 and 4 Fe2 spins via Mn-O-Fe bridges, and frustration of both of these interactions leads to an almost perpendicular alignment of the Mn moments at 75 K, although Fe1 spin canting occurs such that their components in the $b$-direction are antiferromagnetically coupled to the Mn spins. Further canting occurs at 5 K as the Mn spins become more fully ordered and Fe1 and Fe2 spins cant out of the $bc$-plane. This magnetic order breaks the mirror-plane symmetries of the lattice perpendicular to the $a$ and $c$ axes, so an exchange-strictive distortion might be expected but is not observed within the resolution of the present data.

The possibility of internal electronic distortions was investigated by using the Bond Valence Sum (BVS) method to estimate oxidation states for the Fe sites via a standard interpolation method with bond distances derived from the neutron refinements. The BVS results in Figure 4c show that both Fe1 and Fe2 sites have mixed Fe$^{2+}$/Fe$^{3+}$ charge states at 75 – 400 K, but charge ordering is evident at 5 K with Fe1 and Fe2 respectively tending to Fe$^{3+}$ and Fe$^{2+}$ states. The effect of Jahn Teller distortion (QT) is also calculated for both Fe site. Charge localisation as Fe$^{2+}$ is expected to lead to Jahn Teller compression of the Fe2O$_6$ octahedron (negative values of the QT parameter reported in ref. 5), and although no large changes are observed on cooling below 75 K, the negative QT for Fe2 (Fe$^{2+}$) and near zero value for Fe1 (Fe$^{3+}$) corroborate the BVS charge ordering results. Hence a Fe$^{2+}$/Fe$^{3+}$ charge ordering transition occurs at $T_{CO}$ = 60 K in MnFe$_2$O$_5$, with concomitant spin canting. As the Fe$^{3+}$ and Fe$^{2+}$ states respectively localise at inequivalent Fe2 and Fe1 sites there is no symmetry-breaking distortion associated with the charge order, or with the Fe$^{2+}$ orbital order, although the associated 5 K spin order does break the Cmcm lattice symmetry as noted above.

Our previous study of the crystal structure using powder synchrotron X-ray diffraction data showed that anisotropic thermal expansion of the lattice parameters is observed over the temperature range 90-400 K. Further data collected between 15 and 120 K here confirm that the orthorhombic Cmcm structure persists to low temperatures. A discontinuity in the refined lattice parameters and cell volume is observed at 60 K, (Fig. 5) corresponding to the divergence in the ZFC and FC magnetisation measurements. This likely marks the onset of the charge ordering observed in the 5 K but not the 75 K neutron diffraction data.

**Electrical properties**

Electrical resistivity measurement of a polycrystalline pellet of MnFe$_2$O$_5$ shows semiconducting behaviour, with the resistivity increasing when cooled (Fig. 6). The sample resistance was too great to be measured below 260 K. Two linear regions are observed in the inset plot of log(resistivity) vs. inverse temperature, with a change of slope near 285 K. Fitting the Arrhenius equation $\rho = A\exp(E_a/kT)$ to the two linear regions gives the activation energy $E_a$ as 210 meV above and 280 meV below the 285 K crossover. This temperature does not match any of the observed magnetic transition temperatures, and may correspond to a change from defect-dominated to intrinsic bandgap conduction.

![Diagram](https://example.com/diagram.png)
Fig. 4 (a) Magnetic structures of MnFe$_3$O$_5$ at 5, 75 and 300 K. The network of Fe$_6$ octahedra is shown, with Mn$^{2+}$ in trigonal prismatic sites within channels parallel to the $a$-axis. (b) The temperature evolution of the ordered Mn, Fe1 and Fe2 magnetic moments. (c) Temperature evolution of BVS and $Q_{JT}$, with closed/open symbols representing Fe1/Fe2 sites.

Fig. 5 Changes in the lattice parameters and cell volume obtained from powder synchrotron X-ray diffraction experiments.

Fig. 6 Log of electrical resistivity of MnFe$_3$O$_5$ measured between 260 and 380 K, with insert showing the plot against reciprocal temperature.

The presence of structural channels within a framework of redox-active FeO$_6$ octahedra, and moderate electrical conductivity, suggests that MnFe$_3$O$_5$ would be worth investigating as a battery electrode material. Full reduction of Fe$^{3+}$ to Fe$^{2+}$ through lithium insertion would give Li$_2$MnFe$_3$O$_5$, and as cycling between this and the parent phase does not involve oxidation of Mn$^{3+}$ then structural degradation from formation of Jahn-Teller active Mn$^{3+}$ is avoided. However, alternative methods to high pressure synthesis are likely to be needed to make sufficient quantities of MnFe$_3$O$_5$ for practical battery research.

Conclusions

MnFe$_3$O$_5$ is found to have a rich magnetic ordered behaviour between 5 and 400 K. On cooling from high temperatures, the moments at the iron sites order with an antiferromagnetic arrangement at $T_N = 350$ K, but additional order of Mn moments below $T_{Mn} = 150$ K leads to canting of Fe1 spins and a significant ferrimagnetic moment. A similar canting of Fe (and Re) spins due to low temperature order of Mn$^{3+}$ moments was reported in the double perovskite Mn$_2$FeReO$_6$.\textsuperscript{15} Fe$^{2+}$/Fe$^{3+}$ charge ordering drives a further transition at $T_{CO} = 60$ K with additional canting of all spins. Similar charge and orbital orders are observed in Fe$_3$O$_4$ and Fe$_5$O$_8$. Electrical resistivity measurements reveal semiconducting behaviour in MnFe$_3$O$_5$ and a small change in the activation energy at 285 K may correspond to a crossover from defect to intrinsic conduction regimes. MnFe$_3$O$_5$ is worth further investigation as a potential lithium battery electrode material.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements
We acknowledge financial support from European Research Council (ERC), Engineering and Physical Sciences Research Council (EPSRC) and Science, Technology Facilities Council (STFC) and the ESRF for provision of beamtime. We would also like to thank Andy Fitch (ESRF) and James Cumby, Alexander J. Browne, Giuditta Perversi and Paul M. Sarte (Edinburgh) for assistance provided.

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