Ferri- and ferro-magnetism in CaMnMReO6 double double perovskites of late transition metals M = Co and Ni

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Two new double double perovskites of ideal compositions CaMnReO₆ (M = Co, Ni) are reported. The M = Co material has refined composition CaMnₓCo₁₋ₓReO₆ and orders ferrimagnetically below Tₑ = 188 K with a relatively large saturated magnetisation of 4.5 μB. The M = Ni product, CaMnₓNi₁₋ₓReO₆, is a remarkable example of a ferromagnetic oxide with four distinct spin sublattices all collinearly ordered below Tₑ = 152 K. The versatility of the perovskite structure has led to the synthesis and study of many perovskite materials exhibiting a wide range of useful properties. 1:1 ordering of cations at the A and B site cations in the basic ABO₃ perovskite structure in layered, columnar or rock-salt arrangements leads to AA’B₂O₆ and AB’O₆ double perovskites (DPv) but combinations of cation order in both sublattices are uncommon. These double double (or doubly ordered) perovskites usually combine layered and rock-salt motifs of A and B cations respectively. A new double double perovskite (DDPv) type was recently reported to combine a columnar arrangement of A-site cations with the rock-salt order of B and B’ cations for RMnMnOsO₆ (R = rare earth). This structure crystallizes with the tetragonal P₄/n space group and has five independent cation sites according to the composition AA’A’₂B’₂B’O₆: A (Wyckoff position 4e), A’ (2a) and A’’ (2b) sites with ten-fold, tetrahedral and square planar coordinations, respectively, and octahedral sites for B (4c) and B’ (4d) cations. CaMnReO₆ double double perovskites with the same structure were subsequently reported for M = Mn and Fe. These are found to have different charge distributions and magnetic properties, as CaMnFe₂ReO₆ is ferrimagnetic with Curie temperature Tₑ = 500 K, while CaMnMn²ReO₆ has a mix of ferro- and antiferro-magnetic spin sublattices ordered below 120 K.

Samples of ideal composition CaMnCoReO₆ and CaMnNiReO₆ have been treated under high pressure and high temperature conditions using a Walker-type multi anvil apparatus. A stoichiometric mixture of CaMnO₃ (obtained by heating a pellet of CaCO₃ and MnO in a 1:1 ratio at 1473 K for 24h under an O₂ flow with intermediate grinding and repelletising), ReO₂ and MO (M = Co, Ni) oxides, were packed in a Pt capsule and treated at 10 GPa and 1373 K (for M = Co) or 1473 K (M = Ni) during 20 minutes. The samples were quenched and slowly depressurized to ambient conditions. These synthesis temperatures were found to be optimal for production of the DDPv type phases, as syntheses at 1423-1473 K for M = Co lead to secondary Re phases, while products made at 1373-1423 K for M = Ni contained unreacted ReO₂ and NiO. However secondary phases were present for both materials and were not eliminated in repeated syntheses. Laboratory X-ray powder diffraction (XPD) patterns were collected using a D2 Bruker diffractometer. Magnetic susceptibilities (ZFC-FC under an applied magnetic field of 1000 Oe) and magnetization-field loops up to 7 T were measured using a MPMS SQUID magnetometer. Neutron powder diffraction (NPD) patterns collected at room temperature and 1.5 K on the WISH beamline at the ISIS facility were used to determine crystal and magnetic structures.

Fits to XPD (supplementary Fig. S1) and NPD data showed that both CaMnMReO₆ (M = Co, Ni) samples adopt the P₄/n DDPv structure. XPD patterns reveal traces of secondary ReO₂ (1.7 and 3.0 weight% respectively for M = Co, Ni). The NPD patterns (Figure 1 and ESI) show the presence of a rocksalt type monoxide impurity (8.0 and 17.1% respectively for M = Co, Ni). In both cases the cation distribution in the latter refined to composition M₀.₆₂Mn₀.₃₈O₃, suggesting that high pressure...
stabilises a rocksalt-related M₂MnO₃ phase, but possible long range cation ordering was not detected in this study. Table 1 summarizes the refined DDPv structures of the CaMnMReO₆ (M = Co, Ni) compounds.

The varied neutron scattering lengths for the metals in these compounds (4.70, -3.73, 2.49, 10.3, 9.2 fm for Ca, Mn, Co, Ni and Re respectively) provide a high degree of contrast enabling cation mixing to be investigated across the five metal sites in the AA’BB’O₆ structure. The A = Ca (ten-fold) and B’ = Re (octahedral) sites were found to be fully occupied in both materials within experimental error in the NPD refinements. CaMnCoReO₆ has 96% Co at the octahedral B site, but 30-40% of Co substitutes for Mn at the A’ (tetrahedral) and A” (square planar) sites. This leads to an overall Co-rich composition CaMn₃Co₂ReO₆ relative to the ideal formula. CaMnNiReO₆ has a mix of 37% Mn and 63% Ni at the B site but with 10-20% Ni substituting at the A’ and A” sites, leading to an overall Ni-poor composition CaMn₂Ni₀.₅ReO₆. This is consistent with the higher proportion of observed M₂MnO₃ secondary phase than in the M = Co sample. Apparent lack of Ca in the products may be because this has formed amorphous phases, or crystalline products below limits of XPD detection. XPD data were fitted using the NPD structural models, and with M/Re disorder at the B-sites refined. Small antisite occupancies of 3.4(1) and 2.5(1)% for M = Co and Ni respectively show that the high charge contrast between (Mn/Co/Ni)²⁺ and Re⁶⁺ leads to a high degree of B-site ordering.

The cation oxidation states at each atomic position have been estimated from BVS calculations, as shown in Table 1, confirming that the formal charge distributions are Ca²⁺A’²⁺₀.₄₆A”²⁻₀.₅B²⁺Re⁶⁺O₆ in both cases with A’, A”, and B sites occupied by mixtures of Mn²⁺ and Co²⁺/Ni²⁺. This disorder reflects the similarity of ionic radii between the divalent transition metal cations. However, no substitutions at the Ca and Re sites were found in either material.

Table 1. Sites showing Wyckoff labels, atomic coordinates, occupancies and BVS values resulting from the Rietveld fits of the room temperature NPD patterns of CaMnCoReO₆ (upper lines) and CaMnNiReO₆ (lower lines). The space group is P4₂/n and cell parameters are a = 7.666(1) and 7.6868(4) Å and c = 7.6899(4) and 7.693(1) Å respectively. Biso values, constrained for cations and oxygen sites, are 0.2(1) and 1.0(1) Å² for M = Co and 0.8(1) and 1.7(1) Å² for M = Ni respectively. Residuals; $R_p = 2.24\%$, $R_w = 2.08\%$, $R_I = 5.63\%$, $R_B = 4.69\%$ and $R_R = 2.59\%$, $R_w = 3.19\%$, $R_I = 7.09\%$, $R_B = 5.28\%$ respectively.

<table>
<thead>
<tr>
<th>Site</th>
<th>x</th>
<th>y</th>
<th>z</th>
<th>Mn/M occ</th>
<th>BVS</th>
</tr>
</thead>
<tbody>
<tr>
<td>A’ (2a)</td>
<td>0.75</td>
<td>0.75</td>
<td>0.75</td>
<td>0.692/0.308(1)</td>
<td>1.9</td>
</tr>
<tr>
<td>A” (2b)</td>
<td>0.25</td>
<td>0.25</td>
<td>0.75</td>
<td>0.612/0.388(1)</td>
<td>1.7</td>
</tr>
<tr>
<td>Ca (4e)</td>
<td>0.25</td>
<td>0.75</td>
<td>0.779(1)</td>
<td>0.783(1)</td>
<td>2.0</td>
</tr>
<tr>
<td>B (4c)</td>
<td>0</td>
<td>0.5</td>
<td>0.5</td>
<td>0.040/0.960(1)</td>
<td>2.0</td>
</tr>
<tr>
<td>Re (4d)</td>
<td>0</td>
<td>0</td>
<td>0.5</td>
<td>0.366/0.634(1)</td>
<td>1.7</td>
</tr>
<tr>
<td>O₁ (8g)</td>
<td>-0.055(2)</td>
<td>0.556(2)</td>
<td>0.239(1)</td>
<td>1.9</td>
<td></td>
</tr>
<tr>
<td>O₂ (8g)</td>
<td>-0.049(1)</td>
<td>0.559(2)</td>
<td>0.24(1)</td>
<td>1.9</td>
<td></td>
</tr>
<tr>
<td>O₃ (8g)</td>
<td>-0.237(2)</td>
<td>-0.049(1)</td>
<td>0.571(1)</td>
<td>1.9</td>
<td></td>
</tr>
<tr>
<td>O₄ (8g)</td>
<td>-0.240(3)</td>
<td>-0.049(1)</td>
<td>0.566(1)</td>
<td>1.9</td>
<td></td>
</tr>
<tr>
<td>O₅ (8g)</td>
<td>-0.263(1)</td>
<td>0.061(1)</td>
<td>-0.0298(5)</td>
<td>1.9</td>
<td></td>
</tr>
<tr>
<td>O₆ (8g)</td>
<td>-0.265(2)</td>
<td>0.055(1)</td>
<td>-0.029(1)</td>
<td>1.9</td>
<td></td>
</tr>
</tbody>
</table>

Magnetic susceptibility and field dependent magnetization measurements are depicted in Figure 2. CaMnCoReO₆ and CaMnNiReO₆ show ferrimagnetic behaviour below $T_C$ = 188 and 152 K respectively. A fit of the Curie-Weiss law to the high temperature inverse susceptibilities (shown in ESI) results in Weiss constants $\theta = 188$ and 151 K respectively, indicative of ferromagnetic exchange interactions for both compounds. The effective paramagnetic moments are 5.8 and 7.2 $\mu_B$/f.u. for M = Co and Ni respectively. Predicted values are 6.9 and 7.2 $\mu_B$/f.u. for the refined compositions CaMn₂Co₁₃ReO₆ and CaMn₁₂Ni₀₈ReO₆, assuming all cations are in high spin states. Both experimental values lie below the predicted values which most likely reflects the presence of impurity contributions to the susceptibilities. Low temperature $M-H$ loops in Fig.2 show that both materials have substantial saturated magnetic moments, of 4.5 and 1.8 $\mu_B$/f.u. for M = Co and Ni respectively, indicating that they have ferri- or ferro-magnetic orders at low temperature. Their coercivities are small and the
M = Co hysteresis loop shows a narrowing near \( H = 0 \) indicating that competing ferro- and antiferro- magnetic exchange interactions are present.

FIGURE 2. Magnetic susceptibilities of CaMnCoReO\(_6\) (top) and CaMnNiReO\(_6\) (bottom) with insets showing magnetization-field loops at 7 and 2 K respectively.

Table 2. Observed saturated magnetization at low temperatures \( M_{sat} \) and NPD refined magnetic moments for tetrahedral A', square planar A", and octahedral B and B' sites, and the resulting net magnetic moment per formula unit \( \mu \), for CaMnMReO\(_6\) with \( M = \text{Co/Ni} \). All quantities are in \( \mu_\text{B} \) units.

<table>
<thead>
<tr>
<th></th>
<th>( M_{sat} )</th>
<th>( \mu(\text{A}') )</th>
<th>( \mu(\text{A''}) )</th>
<th>( \mu(\text{B}) )</th>
<th>( \mu(\text{B'}) )</th>
<th>( \mu )</th>
</tr>
</thead>
<tbody>
<tr>
<td>Co</td>
<td>4.5</td>
<td>2.9(1)</td>
<td>-1.3(1)</td>
<td>3.27(3)</td>
<td>0.3(1)</td>
<td>4.4</td>
</tr>
<tr>
<td>Ni</td>
<td>1.8</td>
<td>3.0(1)</td>
<td>1.2(2)</td>
<td>1.2(2)</td>
<td>0.2(1)</td>
<td>3.5</td>
</tr>
</tbody>
</table>

Magnetic diffraction peaks observed in the 1.5 K NPD patterns of CaMnCoReO\(_6\) and CaMnNiReO\(_6\) are indexed by propagation vector \([0 0 0]\). Rietveld fits \( R_{\text{mag}} = 5.54 \% \) and 5.83 \% respectively yield the magnetic structures shown in Fig. 3. Both CaMnCoReO\(_6\) and CaMnNiReO\(_6\) show a simultaneous order of A and B sublattices with spins along the z axis. All of the magnetic moments were refined independently, converging to a net ferrimagnetic arrangement of A and B sublattices for CaMnCoReO\(_6\) and their ferromagnetic alignment for CaMnNiReO\(_6\), as summarised in Table 2. Other magnetic modes were not consistent with the data. See ESI for further details of the magnetic symmetry analysis. An additional magnetic diffraction peak was assigned and fitted as the \((\frac{1}{2}\frac{1}{2}\frac{1}{2})\) peak of the Mn\(_{0.67}\)Mn\(_{0.33}\)O impurity phases. This peak is also observed in the 300 K pattern of CaMnNiReO\(_6\). The high ordering temperature is consistent with the Néel temperatures of NiO \((T_N = 523 \text{ K})\) and MnO \((T_N = 122 \text{ K})\), so a transition near 390 K may be interpolated for Mn\(_{0.67}\)Mn\(_{0.33}\)O. An equivalent interpolation based on \( T_N = 291 \text{ K} \) for CoO predicts spin ordering around 235 K for Co\(_{0.67}\)Mn\(_{0.33}\)O, consistent with the observed appearance of magnetic neutron diffraction peaks between 300 and 1.5 K.

FIGURE 3. Crystal and magnetic structures of CaMnCoReO\(_6\) (left) and CaMnNiReO\(_6\) (right) with the z-axis vertical. Orange, red, yellow, purple and blue spheres represent Ca, A', A", B and Re sites respectively. Polyhedra show the tetrahedral, square planar and octahedral coordination of A', A" and B sites, and arrows show the spin directions scaled to the refined magnitudes of the ordered moments.

It is notable that all four magnetic sublattices in these two double double perovskites order ferromagnetically. The Re moments are very small, consistent with those in other Re\(^{11+}\) oxides\(^{16}\), but refine to be parallel to the other B sublattice spins whereas those in CaMnFe\(^{3+}\)Re\(^{5+}\)O\(_6\) are antiparallel due to spin polarized conduction that leads to a high Curie temperature of 500 K. The \( T_c \)'s below 200 K and parallel alignment of B = Co, Ni and B' = Re spins in the CaMnMReO\(_6\) \((M = \text{Co, Ni})\) materials indicates that they are insulating ferromagnets, although we have not yet obtained well-sintered ceramic pellets for direct conductivity measurements to confirm this. Ordered tetrahedral A' site moments are larger than those at square planar A" sites in both materials, reflecting the greater degree of cation disorder at the latter sites. B site moments are relatively large \((3.3 \mu_\text{B})\) for the \( M = \text{Co}\) material which has almost no cation disorder at this site, whereas substantial Mn/Ni mixing leads to a much smaller ordered moment of 1.2 \( \mu_\text{B} \) for \( M = \text{Ni}\). The reduction of the site moments compared to ideal 2S values reflects magnetic
disorder resulting from a complex mix of antiferro- and ferromagnetic exchange interactions between the different sites and cation d\(^{6}\) configurations. This may be sensitive to the precise compositions of the studies samples, both of which are off-stoichiometric as noted above.

\(\text{CaMnCoReO}_6\) is a ferrimagnet overall as the ferromagnetic \(A''\) sublattice spins are antiparallel to those of the \(A', B\) and \(B'\) sublattices. The net moment of \(4.4 \mu_\text{B}\) for \(M = \text{Co}\) predicted from the NPD results is in good agreement with the observed net magnetization from the hysteresis loop (4.5 \(\mu_\text{B}\) at 7 K). The refined spin structure of \(\text{CaMnNiReO}_6\) is remarkable as all four ferromagnetic sublattices are parallel so this material is a rare example of a ferromagnetic (and likely insulating) oxide. The observed magnetization is below the predicted value from the refined NPD moments, most likely reflecting the presence of the antiferromagnetic \(\text{NiO}_6\text{MnO}_3\text{ReO}_6\) impurity phase. Ferrimagnetism could also account for the smaller saturated moment, but attempts to refine such models against the neutron data always resulted in the ferromagnetic spin structure shown.

In conclusion, two new members of the double double perovskite family, of ideal compositions \(\text{CaMnCoReO}_6\) and \(\text{CaMnNiReO}_6\), have been synthesized under high pressure and high temperature conditions. They retain the cation ordering pattern of \(\text{CaMnFeReO}_6\), but with more substantial Mn/M cation mixing across three of the five available cation sites leading to non-stoichiometric \(\text{CaMn}_{x/3}\text{Co}_{x/3}\text{ReO}_6\) and \(\text{CaMn}_{x/2}\text{NiO}_6\text{ReO}_6\) compositions. Ferromagnetic ordering within all of their spin sublattices occurs at a single magnetic transition, unlike previous studied \(M = \text{Mn}\) and \(\text{Fe}\) analogues where two transitions were observed. \(\text{CaMnCoReO}_6\) orders ferrimagnetically below \(T_c = 188\) K with a relatively large saturated magnetization of 4.5 \(\mu_\text{B}\). \(\text{CaMnNiReO}_6\) is a remarkable example of a ferromagnetic oxide with four distinct spin sublattices all collinearly ordered below \(T_c = 152\) K.

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Conflicts of interest

There are no conflicts to declare.

Notes and references